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# The effect of solvent of titanium precursor in the sol-gel process on the activity of $TiO_2$ nanoparticles for $H_2$ production

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Article Information	Abstract			
Article History:	A modified sol-gel process has been found to significantly improve the photocatalytic			
Received: 31 July 2017 Received in revised form: 24 Oct 2017 Accepted: 04 Nov 2017	activity of $TiO_2$ nanoparticle in the process of solar hydrogen production. The surface of $TiO_2$ nanoparticles were modified by the optimization of solvent of titanium precursor (acetic acid and/or ethanol) in the sol-gel method. A multi technique approach (SEM, XRD, FTIR, UV-DRS and TGA) was used to characterize the prepared $TiO_2$ nanoparticles. The photocatalytic hydrogen production was tested using a suspension of photocatalyst $TiO_3$ at 10 vol. % methanol under natural			
Keywords	solar light. The produced hydrogen was subjected to gas chromatography with			
$H_2$ production Solar light TiO <sub>2</sub> nanoparticles Photocatalyst	a continuous flow of $N_2$ in the photoreactor system. It was found that the $TiO_2$ nanoparticles synthesized with acetic acid as the solvent of titanium precursor, $TiO_2$ -AA, have a better photocatalytic activity for hydrogen production compared to nanoparticles synthesized with ethanol, $TiO_2$ -EA. The obtained results showed that the better crystallinity, small size and proper surface properties of $TiO_2$ -AA nanoparticles is due to higher photoactivity.			

# 1. Introduction

Hydrogen has high energy efficiency without any environmental pollution [1, 2]. Nowadays, hydrogen

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is obtained from steam reformation of natural gas, liquefied hydrocarbons, and liquefied petroleum gas [3, 4]. Although, industrial production of hydrogen consumes high level of energy and produces huge

amount of carbon dioxide as by-product and greenhouse gases [1]. The photocatalytic splitting of water under sunlight for producing hydrogen is one of the potential methods for getting clean, low cost and eco-friendly fuel [4]. Titanium dioxide as a photocatalyst has been the subject of interest for many researchers because of its several advantages such as non-toxicity, eco-friendly feature, availability and chemical stability. Titanium dioxide exists in nature in three phases; rutile, anatase and brookite [3, 5]. Photocatalytic activity of titanium dioxide depends on its crystal structure, degree of crystallinity and specific surface. The anatase phase has more photocatalytic activity than the other two phases due to its greater crystallinity and surface, and smaller particle size [6,7]. There are several methods for synthesizing nanoparticles such as sol-gel, hydrothermal, precipitation and flame synthesis. The morphology, structure and photocatalytic activity are dependent on the method of synthesizing nanoparticles [8–10]. Because of a wide range of advantages, the sol-gel method has attracted the focus of many researchers. Some of the notable advantages includes high chemical purity, high homogeneity, fine-scale and controllable morphology [6, 11]. In this method, the precursor molecule (usually metal alkoxides) is hydrolyzed in a solvent and a colloidal dispersion of sol is formed. The accumulation of sol particles creates an infinite network of particles which consequently leads to the gel production. The resulting gel is dried and then powdered and the obtained powder is heated for calcination [6]. Alkoxides precursors, such as titanium butoxide and titanium isopropoxide, are usually used to produce titanium dioxide nanoparticles, and the precursor is generally dissolved in an alcoholic solvent (for instance ethanol) [12, 13]. Making changes in the synthesis method to obtain TiO<sub>2</sub> nanoparticles with more suitable properties have been reported in multiple research. In 2014 Xin and coworkers synthesized TiO<sub>2</sub> nanoparticles using the hydrothermal method and examined the effect of calcination temperature on the material's properties and its photocatalytic activities in producing Hydrogen [15]. R. Vijayalakshmi and

V. Rajendran compared two methods, hydrothermal and sol-gel, for synthesizing TiO<sub>2</sub> nanoparticles in the same ambient conditions, they reported that the synthesized particles using the sol-gel method have more crystallinity and smaller size than the particles synthesized using the hydrothermal method [16]. Loryuenyong and coworkers synthesized TiO<sub>2</sub> nanoparticles using the sol-gel method with isopropanol as solvent and active carbon as templates, they reported an increase in the specific surface of the synthesized TiO<sub>2</sub> nanoparticles and an increase in the band gap (0.3 to 0.6 eV) that could be due to the effect of restricted movement of electrons [17]. Lou and coworkers also synthesized TiO<sub>2</sub> using the sol gel method with the help of polyethylene glycol (PEG) and cetytrimethy lammonium bromide (CTAB) and examined the effect of their presence on the size of the particles [18]. Dinh and coworkers have synthesizied TiO<sub>2</sub> using the solvothermal method using water as the agent of hydrolysis and oleic acid (OA) and oleylamine (OM) as two distinct capping surfactants, and by changing this factor they examined their effect on the crystallinity and the shape of the particles [19]. Sanaz Naghibi and coworkers used the Taguchi method have examined the different factors, such as pH and precursor of Ti, they used the solgel and then hydrothermal method for synthesizing nanoparticles with the highest crystallinity and smallest particle size .[20] One common method for synthesizing nanoparticles is the sol-gel method with the help of an organic solvent. In 2015, Zhang and partners synthesized TiO<sub>2</sub> nanoparticles using the sol- gel method with the use of DMAC as a solvent and glacial acetic acid as a catalyst and researched the effect of the ratio between the solvent and TBT amount [21]. In sol-gel processes, the morphology of the final product is strongly influenced by the reactivity of the precursor [14]. So it is an important subject to consider in the details of this method, since the details play an important role in the size of particles and photocatalytic activity of TiO<sub>2</sub> in processes such as water splitting.

In the present work,  $TiO_2$  nanoparticles were synthesized via the sol-gel route with two different

solvent of Ti precursor, acidic solvent and alcoholic solvent. We present the effect of the solvent of titanium precursor on the characterizations of  $\text{TiO}_2$  nanoparticles. The photocatalytic activity of  $\text{TiO}_2$  was investigated by water splitting and the production of H<sub>2</sub> under natural solar light.

## 2. Experimental

## 2-1. Preparation of TiO<sub>2</sub> nanoparticles

All the chemical materials were purchased from Merck and were used without any further purification. The  $\text{TiO}_2$  nanoparticles were synthesized by the solgel method in solvents of glacial acetic acid (AA) and/or anhydrous ethyl alcohol (EA) as a precursor solvent. The titanium tetraisopropoxide (TTIP) was used as the precursor of titanium. The prepared TiO<sub>2</sub> nanoparticles in solvents of glacial acetic acid (AA) and anhydrous ethyl alcohol (EA) were named as TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA, respectively.

experimental In an route, titanium the tetraisopropoxide as precursor (8 ml TTIP) was solved in 40 ml of absolute ethanol and /or acetic acid in room temperature. Forced hydrolysis of the TTIP solution was achieved by adding a certain volume of deionized water (40 ml). Subsequently, the solutions were ultrasonicated (Modle; Hielscher UP400S) for 15 min and then the stirring process was continued for a further 30 min, the solutions were again ultrasonicated for 15 min until a clear solution was formed. The prepared solutions were kept in an oven at 70 °C for 16 h. The gels of TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA were dried at 120 and 100 °C, respectively. The prepared powders were crushed well and calcinated in air in the muffle furnace for 2 h at different temperatures. The as-synthesized samples were denoted as TiO<sub>2</sub>-AA-X and TiO<sub>2</sub>-EA-X, where x refers to the calcination temperature.

## 2.2. Catalyst characterization

Thermal Gravimetric Analysis (TGA) was carried out

for the titanium dioxide samples using simultaneous TGA, model of Perkin-Elmer, and STA-6000 under Ar atmosphere with a heating rate of 10 °C.min<sup>-1</sup>. The XRD patterns were recorded on a Philips X'pert Pro MPD model using Cu Ka radiation as the X-ray source. The diffraction pattern was taken in the  $2\theta$  range  $10-80^\circ$ . The average crystallite size of the anatase phase was determined according to the Scherrer equation. The morphology and size of the nanoparticles were characterized using a scanning electron microscope (SEM, Philips XL-30ESM). Fourier transform infrared spectra (FT-IR) was measured by a JASCO 6300 spectrophotometer (200-4000 cm<sup>-1</sup>). UV-Vis diffuse reflectance spectra (DRS) were determined by a UV-Vis Spectrophotometer (JASCO, V-670).

### 2.3. Photocatalytic hydrogen production

Photocatalytic hydrogen evolution experiments were carried out in a self-designed system under natural solar light as shown in Fig. 1. The reaction was typically performed in a pyrex reactor by adding 0.2 g catalyst to a solution of deionized water (270 ml) and methanol (30 ml) under stirred condition. The reactor was then sealed and before the irradiation, the mixture was degased with nitrogen gas for about 30 min to completely remove any oxygen. A nitrogen gas flow (100 mL/min) was passed through the mixture during the irradiation. A magnetic stirrer was used for uniform dispersion of the catalysts in the reactor during solar irradiation. The amount of H<sub>2</sub> produced was measured using online gas chromatography (Agilent 6890N) equipped with an injection valve (VALCO type), a molsieve 5A column, a TCD detector and N<sub>2</sub> as carrier gas. The calibration plot was obtained by using GC with high purity H<sub>2</sub> and N<sub>2</sub> gases and injected with different concentrations. The results of blank tests showed that the presence of photocatalyst, solar light and scavenger are essential for H<sub>2</sub> generation. For comparison, the  $H_2$  evolution rate using the commercial photocatalyst Degussa P25 was also determined following the same procedure described



Fig. 1. Schematic representation of experimental setup for photocatalytic water splitting under solar irradiation.

above. The photocatalytic hydrogen production under solar light has been investigated at 12:00 am to 3:00 pm in the summer season in Esfahan, Iran  $(32^{\circ}5'N, 51^{\circ}43'E)$ .

# **3. Results and discussion**

### **3.1.** Characterization of nano-TiO<sub>2</sub>

TGA/DTGA profile of representative TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA samples are shown in Fig. 2. The total of mass loss is 21% for  $TiO_2$ -AA and 11% for  $TiO_2$ -EA. The TGA curve (Fig. 2 a) shows that the mass loss for TiO<sub>2</sub>-AA occurs in two different stages. The first step of mass loss (5.4 %) is in the temperature range of 50 to 250 °C, which is related to the dehydration and evaporation of organic substances [22, 23]. The second step of mass loss (13.9%) is seen in the range of 250 to 410 °C that is assigned to the burnout of hydroxyl groups and the organic substances [22, 24]. The TGA curve of TiO<sub>2</sub>-EA is shown in Fig.2b. The TiO<sub>2</sub>-EA does not have a separate decomposition temperature. Instead, it exhibits a gradual mass loss (11%) from 50 to 700 °C. The TG curve of  $TiO_2$ -EA shows an initial mass loss of ~6% at nearly 50 °C and continuing until 200°C attributable to physisorbed water and ethanol solvents [25, 26]. In the range of 200 to 420 °C, the second step of mass loss (3.6%) that is related to the removal of hydroxyl groups and other organic groups [24] is seen with a peak of DTGA at around 397 °C. The DTGA curve of  $\text{TiO}_2$ -AA clearly shows the two peaks with different heights at around 309.34 and 364.45 °C. These peaks are related to the decomposition of organic compounds which participate in the synthesis [22].

X-ray diffraction (XRD) is one of the oldest and most useful approaches in identifying crystalline materials [27]. XRD analysis was carried out in the range of  $2\theta$ =10–80 for samples to investigate the structural changes of TiO<sub>2</sub> affected by synthesis methods, as demonstrated in Fig. 3. The synthesized nanoparticles showed crystalline nature with the pure anatase phase of TiO<sub>2</sub>. The diffraction peaks at 25.44°, 37.95°, 48.21°, 54.034°, 55.22°, 62.82°, 68.91°, 70.46° and 75.23° correspond to the crystal planes [101], [004], [200], [105], [211], [204], [220], [220] and [215], respectively; this is attributed to reflections of anatase TiO<sub>2</sub> which is compared with JCPD Card No.(21-1272) [16, 28].

The average crystallite size of the synthesized TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA were calculated by using Scherrer's equation. According to Scherrer's formula, (D = 0.9  $\lambda$ / ( $\beta \cos\theta$ ) where D is the average crystallite size (nm),  $\lambda$  is the applied X- ray wavelength ( $\lambda$  =0.15406 nm),  $\theta$  is the diffraction angle and  $\beta$  is the full width at half maximum (FWHM) [16]. The size of nanoparticles from the strongest peak (at 2 $\theta$  of 250) was calculated [29]. According to this formula, the average crystallite sizes of TiO<sub>2</sub>-AA-450 and TiO<sub>2</sub>-EA-450



Fig. 2. TGA/DTGA curves of a) TiO<sub>2</sub>-AA, b) TiO<sub>2</sub>-EA.



Fig.3. The XRD pattern of TiO<sub>2</sub>-AA-450 and TiO<sub>2</sub>-EA-450 samples.

samples are 14.1 and 17 nm, respectively. The optimum particle size corresponding to the highest photoactivity in titanium dioxide nanoparticles systems is 10–14 nm [30]. The lattice parameters

were determined using these equations:  $d_{(hkl)} = \lambda/2\sin\theta$ ,  $1/d_{(hkl)} = (h^2+k^2)/a^2 + 12/c^2$ , where  $d_{(hkl)}$  is the distance between crystal planes of (hkl),  $\lambda$ is the X-ray wavelength,  $\theta$  is the diffraction angle of crystal plane (hkl), hkl is the crystal plane index, and a, b, and c are lattice parameters (in anatase phase of  $TiO_2$ , a = b - c) [5]. The parameters of the lattice for the synthesized nanoparticles have been calculated and represented in Table 1. The lattice parameters of the synthesized samples were compared to the anatase phase (a=3.785A°, C=9.514A°) [31]. Some changes are seen that show the different procedures forming crystalline structures in the samples. Also, a difference in the forming procedure of the crystalline structure was observed in the analysis of the TGA thermogram.

Table 1. The crystallite sizes and lattice parameters of  $TiO_2$ -AA and  $TiO_2$ -EA nanoparticles

Nano particle	FWHM	Particle size (nm)	Lattice parameters	
			(A <sup>0</sup> )	
			a(101)	C(101)
TiO <sub>2</sub> -AA	0.5698	14.1	3.766	9.359
TiO <sub>2</sub> -EA	0.5011	17	3.739	9.064

It is a well-accepted idea that the diffraction peak's intensity points out the crystallization degree of samples [5]. As shown in Fig. 3, the greater the intensity of the  $TiO_2$ -AA peaks suggests the formation of higher crystallinity [32, 33]. High crystallinity, which is a result of a reduction in crystal defects in the structure, can facilitate the separation of photogenerated holes and electrons and leads to a decrease in the recombination chance of charge carriers as well, and so the photocatalytic activity will be increased [34, 35].

Preparing a sample of titanium dioxide nanoparticles in the presence of acetic acid,  $TiO_2$ -AA, permits the process of gel formation to be well-controlled. The presence of additional acetate anion at the surface of  $TiO_2$  particles can prevent the growth of  $TiO_2$ nanoparticles. The existance of this kind of complex of acetate anion at the surface of  $TiO_2$  is likely the reason for the decrease in the particle size of  $TiO_2$ synthesized in the sol-gel method using acetic acid as solvent [36, 37]. The additional acetic acid is not due to the impurities on the surface of  $TiO_2$  after calcinations, which is further confirmed by FTIR spectroscopy. SEM studies were performed to investigate the surface morphology of the synthesized  $TiO_2$ -AA and  $TiO_2$ -EA nanoparticles. Fig. 4 shows SEM images of the samples calcinated at 450 °C. According to Fig. 4, the grains of the particles are spherical, and the  $TiO_2$ -AA nanoparticles were more uniformed than  $TiO_2$ -EA. The SEM images show that the agglomeration degree for the  $TiO_2$ -EA sample is more than the  $TiO_2$ -AA sample and the nanoparticles of  $TiO_2$ -AA have a more uniform morphology and smaller particle size. Further observation indicates that the morphology of the  $TiO_2$ -AA nanoparticles is very rugged and may be beneficial to enhancing the adsorption of reactants due to its high surface area.

To study the role of the precursor solvent on the TiO<sub>2</sub> surface, the FTIR spectra of TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA particles were compared after calcination at 450 °C. As demonstrated in Fig. 5, a broad band at 3400 cm<sup>-1</sup> area can be observed for both samples which indicates asymmetric and symmetric stretching vibrations of the terminating hydroxyl group (–OH) in the surface of synthesized nanoparticles. In addition, the characteristic peak in 1624 cm<sup>-1</sup> is related to the O-H bending vibration of the hydroxyl group in the molecule of the adsorbed water which is observed in both samples [18, 38]. These bands are evidence of chemically and physically adsorbed H<sub>2</sub>O on the surface of synthesized nanoparticles [39].

The characteristic vibrations of the inorganic Ti-O stretch have been observed in the area below 1000 cm<sup>-1</sup> [12]. In the low energy area of the spectrum the bands at 461 cm<sup>-1</sup> are assigned to O-Ti-O and Ti-O-Ti bending vibration, and the broad band at 528 cm<sup>-1</sup> directs to the vibration of the metal-oxygen bond (Ti-O) in the sample of TiO<sub>2</sub> [10, 39, 40]. As shown in Fig. 5a, the IR band related to Ti-O at 528 cm<sup>-1</sup> in the TiO<sub>2</sub>-AA sample move toward a shorter wavelength and transmit to 586 cm<sup>-1</sup>. This result shows that the degree of crystallization in the TiO<sub>2</sub>-AA sample has been increased which is confirmed by the results of the XRD [40]. It declares that the combination between ethanol and Ti precursor become weak, but the combination between acetic acid and Ti precursor become strong, which is in accordance with the



Fig. 4. SEM image of a) TiO,-AA, b) TiO,-EA nanoparticles

TGA results. Therefore, elimination of acetic acid is harder than ethanol and affected the crystallinity of the sample.

FTIR spectrum of adsorbed CO<sub>2</sub> on TiO<sub>2</sub> shows four bands: 1586, 1430, 1317, and 1227 cm<sup>-1</sup>. CO<sub>2</sub> as a Lewis acid can be linked to basic sites with a alkalinity feature in the metal oxides surface. In titanium dioxide with an anatase phase, adsorbed CO<sub>2</sub> produces mainly bidentate carbonate. The surface bidentate carbonate complexes are formed involving the interaction with Ti4+- O2- Lewis acidbase pairs sites in the surface of the anatase phase [41]. These results of the FTIR spectrum suggest that the amount of Ti<sup>4+</sup>-O<sup>2-</sup> pairs on the surface of TiO<sub>2</sub>-AA is more than that on TiO<sub>2</sub>-EA. The spectrum of TiO<sub>2</sub>-EA show an OH peak broader than TiO<sub>2</sub>-AA, while in the spectrum of TiO<sub>2</sub>-AA evidence of adsorb CO<sub>2</sub> is assigned. Adsorption of CO<sub>2</sub> on the surface of TiO, nanoparticles leads to a reduction in surface O-H groups so that the reduction of OH peak intensity in 3400 cm<sup>-1</sup> is identified [42]. As revealed in both spectrums, the organic ligand and the solvent of precursor were completely eliminated after the annealing process. These results match the TGA results which illustrate that at temperatures above 430°C, the mass loss is not observed.

For semiconductor materials, diffuse reflectance spectroscopy (DRS) is an efficient method for characterizing the optical absorption property, which is recognized as one of the important factors for photocatalytic activities [34]. Fig. 6 shows the UV-visible DRS of TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA photocatalysts calcined at 450 °C for 2 h. Compared with TiO<sub>2</sub>-EA, the absorption edge of TiO<sub>2</sub>-AA was shifted towards thelower energy region (i.e. red shift), and the absorption edge was 403 nm. The absorption edge of TiO<sub>2</sub>-EA was about 395 nm. Accordingly, the absorption edges of the modified TiO<sub>2</sub> photocatalysts had been shifted into the high wavelength region, causing narrowing of the band gaps [43]. The minimum energy that is required to transfer an electron from the valance band to conductance band depends upon the band-gap energy Ebg of the photocatalyst and is given as  $E_{h\sigma}(eV) = 1240/\lambda(nm)$ , where  $\lambda$  is the wavelength of the absorption edge in the spectrum [44]. The band gaps for TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA were estimated 3.06 and 3.14 eV, respectively. The enhanced absorption light activity was attributed to the good anatase crystallinity and small crystallite size [45].

#### **3.2.Photocatalytic activity**

The photocatalytic  $H_2$  evolution from water was investigated over TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA samples under natural solar light. The irradiation time dependence on the hydrogen production on TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA samples are shown in Figs. 7 and 8, respectively. All samples prepared by sol-gel and modified sol-gel (acetic acid as solvent) showed the photocatalytic activity for hydrogen production from water. The photocatalytic activity of the H<sub>2</sub>



Fig. 5. FT-IR spectrum of TiO<sub>2</sub> nanoparticles annealed at 450°C a) TiO<sub>2</sub>-AA, b) TiO<sub>2</sub>-EA.



Fig. 6.UV-Vis-DRS of TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA nanoparticles

production system can be explained by using the  $\text{TiO}_2$  photocatalysis mechanism. Under irradiation with energy greater than the band-gap energy of  $\text{TiO}_2$ , the formation of electrons (e-) and holes (h+) in the conduction band and valence band occurred which then can migrate toward the TiO<sub>2</sub> surface. The

resulting hole has strong oxidizing property and the electron is reduced as well. By reducing the absorbed proton in the photocatalyst surface, the electron can produce  $H_2$  which after entering the water phase can be moved to the gas phase. Simultaneously, the hole can be wiped out by the reaction of hole

scavengers such as methanol and ethanol. Hole scavengers prevent the hole's reaction with water and consequently prevents oxygen production, and as a result the possibility of the backward reaction of hydrogen and oxygen and water production is lessened. Likewise, in the effect of the reaction between hole scavengers and hole, the recombination rate of hole and electron is reduced [46, 47].

Methanol as a hole scavenger can react with the holes and oxidized to aldehyde acid and finally  $H_2$  and  $CO_2$ gases are obtained [3]. In order to prove the effect of methanol as a hole scavenger on hydrogen evolution, the hydrogen evolution on the TiO<sub>2</sub>-AA photocatalyst in water without methanol was measured. A trace hydrogen gas, less than the detection limit of our instrument, was seen after 3 h of light irradiation. It can be seen that the amount of the evolved  $H_2$  is increased with the irradiation time for TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA samples as shown in Figs. 7 and 8, respectively. The consequent fall in activity from 450°C to 650°C is due to the combined action of various factors: the lower surface area and the growth of nanoparticle size [22]. The highest activity was observed for TiO<sub>2</sub>-AA-450 with an average hydrogen production rate of 770 (umol/h.g).

The  $H_2$  production activity of the Degussa P25 was compared with TiO<sub>2</sub>-AA and TiO<sub>2</sub>-EA samples (shown in Figs. 7 and 8). The Degussa P25 powder, which is a mixture of the anatase and rutile phases (70–85% anatase, 30–15% rutile), has been used as a standard TiO<sub>2</sub> reference material [48]. Potential of the conduction band of the anatase phase is at a more



Fig.7. The rate of hydrogen evaluation using TiO<sub>2</sub>-AA and P25 samples



Fig. 8. The rate of hydrogen evaluation using TiO<sub>2</sub>-EA and P25 samples

negative position than that of the rutile phase. Thus, the excited electrons easily transfer H<sup>+</sup> ions on the surface of TiO<sub>2</sub> having an anatase phase [49]. The H<sub>2</sub> formation process should be more efficient on the anatse-TiO<sub>2</sub>(101) surface analogous to the rutile-TiO<sub>2</sub>(110) surface [50].

In order to study the effect of the precursor solvent on the photocatalytic activity of the  $\text{TiO}_2$  nanoparticles, hydrogen production rate of the nanoparticles was compared. All synthesized samples have an anatase phase; however, there is a difference in the photocatalytic activity of the particles with the same phase which indicates that in addition to the crystal phase, other factors impact nanoparticles activity [20]. It is believed that the process of producing nano photocatalyst has an important role in the higher activity of produced photocatalyst [51]. We demonstrated that in the present study, the sol-gel method using acetic acid as a solvent of Ti precursor is a better way to synthesize TiO<sub>2</sub> than the sol gel (ethanol) method.

It is obvious that catalytic activity of a solid catalyst largely depends on the catalyst surface [41]. While  $\text{TiO}_2$  surfaces do not create a strong link with molecular hydrogen, various types of adsorption of hydrogen atoms in the surface of  $\text{TiO}_2$  have been reported e.g. bridging hydroxyls, hydride-type H-Ti species and subsurface hydroxyls [52, 53].

After laboratory examinations, Xu et al. presented that molecular hydrogen can be produced in the effect of thermal recombination reaction of hydrogen atoms on bridge hydroxyls. Studies show that bridging hydroxyl species do not have photoactive properties but hydride-type H-Ti species have been known as the photoactive surface particles [52, 54]. CO<sub>2</sub> adsorption on the anatase phase surface indicates that Ti<sup>4+</sup>O<sup>2-</sup> density on the surface of TiO<sub>2</sub>-AA is more than the  $TiO_2$ -EA sample. This is an indicative of a greater percentage of H-Ti on TiO<sub>2</sub>-AA surface, which accordingly confirms more photoactive hydrogen molecule production in the TiO2-AA sample compared to the TiO<sub>2</sub>-EA sample [41]. These results illustrate that the precursor solvent had an effect on the photocatalyst surface and as a result has

an effect on the photocatalytic activity. Furthermore, since a high quality of crystallinity in photocatalytic hydrogen production in water splitting process is preferred over a high surface area, in the conducted examinations  $TiO_2$ -AA samples with a higher quality of crystallinity presented higher hydrogen production [55].

## 4. Conclusion

In sol-gel processes, the final product morphology is strongly influenced by the reactivity of the precursor. In the present study, we demonstrated that acetic acid is a better solvent for athe precursor of Ti to synthesize  $TiO_2$  nanoparticles versus ethanol. The  $TiO_2$  nanoparticles were synthesized with smaller size and higher crystallinity with a selection of acetic acid solvent. These results showed that the precursor solvent affected the surface of the catalyst and the crystallinity of  $TiO_2$  nanoparticles and resulted in higher activity of the photocatalyst for  $H_2$  production.

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